Temperature dependence of microstructure of quenched and aged Al - Zn alloys, Stanko Popović and Željko Skoko, *Physics Department, Faculty of Science, University of Zagreb, 10002 Zagreb, POB.* 331, Croatia; E-mail: spopovic@phy.hr

Keywords: Metal alloys; Microstructure; Phase transitions

Al-Zn alloys are two-phase systems in the equilibrium state: $\alpha(Al-rich fcc, the matrix, M)$ and $\beta(Zn-rich, hcp,$ precipitates). The precipitation and dissolution phenomena and associated phase transitions in a series of alloys up to 62 at% Zn, subjected to various thermal treatments, were studied in situ by XRD [1,2]. Recently, the thermal behaviour of microstructure of the alloys with 44 and 48 at% Zn, was studied in detail. Each alloy was subjected to two thermal treatments: (i) rapid quenching from the solid-solution temperature, T_{SS} , in water at RT (quenching rate >10 4 K/s) and ageing at RT for a week (samples WQ); (ii) cooling slowly from $T_{\rm SS}$ to RT over 5 days and ageing at RT for a week (samples SC). The samples SC were closer to the equilibrium state than the samples WQ; the latter having residual strains, quenched-in vacancies and a non-uniform distribution of $\beta(Zn)$ precipitates. The solid solution, α_{SS} , was formed above 700 K for SC and above 800 K for WO. The rapidly quenched alloys were also aged for 11 months at RT. All the alloys were slowly heated to T_{SS} and slowly cooled to RT, and their microstructure was followed by XRD. As the temperature increased, a decrease of diffraction line intensities took place, due to increased thermal vibrations of atoms. A gradual shift of diffraction lines was noticed due to thermal expansion. The phase $\beta(Zn)$ showed an anisotropy in thermal expansion and a change in the precipitate shape. A partial dissolution of $\beta(Zn)$ in $\alpha(M/\beta)$ took place above 500 K, and a partial transition of $\beta(Zn)$ and $\alpha(M/\beta)$ into α '(fcc, rich in Zn) took place above 550 K. The three phases coexisted up to T_{SS} . In the cooling run the alloys exhibited a temperature hysteresis in relation to the heating run. The microstructure of the samples, aged for 11 months after quenching, was much closer to the equilibrium state than the one of the samples aged for 1 week after quenching. The ideal equilibrium state was not reached in either treatment. A similar behaviour was observed for alloys having 54 and 62 at%Zn [3, 4]. Instead of transitions according to the phase diagram, $\alpha(M/\beta) + \beta - \alpha' + \alpha(M/\alpha') - \alpha_{SS}$, this sequence was found: $\alpha(M/\beta) + \beta - \alpha' + \beta + \alpha(M/\alpha',\beta) - \alpha_{SS}$.

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